

# PATENT SPECIFICATION

DRAWINGS ATTACHED

834739



Date of Application and filing Complete Specification: April 8, 1958.

No. 11111/58.

Application made in Sweden on April 12, 1957.

Application made in Sweden on June 20, 1957.

Application made in Sweden on Feb. 11, 1958.

Complete Specification Published: May 11, 1960.

Index at acceptance:—Classes 22, J(1:4:6:9:13:14:22:28:33); 39(3), H2E4H; 82(1), Y1, Y2(A1:D:G:Q:Z9:Z12); 83(2), A49; and 83(4), R(2:6:10:14B).

International Classification:—B23k, n, p. C04b, H05b.

## COMPLETE SPECIFICATION

### Improvements in or relating to the Manufacture of Electric Resistance Elements

We, AKTIEBOLAGET KANTHAL, a Swedish Company, of Hallstahammar, Sweden, do hereby declare the invention, for which we pray that a patent may be granted to us, and the method by which it is to be performed, to be particularly described in and by the following statement:—

It is known heretofore to butt weld materials in which the basic substance itself is plastically deformable, or ductile, and the temperature necessary to carry out the butt welding operation, as is the case with metals, for example. The characteristic feature in this case resides in that, when the sintering or fusing together is taking place, the actual deformation occurs within the crystal grains, whereby the application of the butt welding method comes quite natural. Further, it is known to be possible in a graphite mold with application of pressure and using externally supplied heat, to sinter together bodies consisting of a material such that plastic deformation cannot be expected to take place within the crystal grains of the basic substance at the temperature at which the sintering operation takes place. In this case the plastic deformation of the material is effected by the crystal grains of the basic substance being displaced in sliding contact with each other, which may be attained only by causing the sintering together to take place under pressure.

The present invention relates to electric resistance elements and to methods of manufacturing same and has for its object in a technically simpler way to produce a reliable welded joint between shaped sintered bodies which have a content of at least 20% by volume of electrically conductive particles, and, for this operation, to make use of the electric conductivity of the material.

In an element according to the invention the sintered bodies are joined by an electric resistance butt welding operation and are

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built-up at least within that zone thereof termed the "weld zone," which is influenced by the welding operation, in part of a basic substance of crystal grains, which *per se* lack or have insufficient plastic deformability and ductility during the butt welding operation, and in part of an additional constituent uniformly distributed in the basic substance and having a softening temperature lower than that of the basic substance, the grain size of the basic substance being at most 20 micra in the welding zone and at most 10 micra beyond this zone.

The statements regarding the grain sizes are in each case understood to concern at least 70% by volume of the basic substance. It has been proved to be impossible, if the grain sizes are too large, to cause the deformations described below, being necessary for reducing the invention into practice. If the grain sizes are excessive the risk is run that there will be ruptures in the weld zone.

In this case, the proportion of basic substance in the material of the welding zone should be at least 20% and at most 99% by volume.

Various types of materials can come into question. In one such type of material the major proportion, suitably more than 65% by volume, of the basic substance consists of  $\text{MoSi}_2$ . In another type of material the basic substance consists of at least 25% by volume of  $\text{MoSi}_2$  and at most 75% by volume of  $\text{Al}_2\text{O}_3$ . The proportion of the aluminium oxide which in the sintering process may combine with the additional constituent is not included in the last-mentioned percentage. Also other substances may, in a manner known *per se*, be included in the basic substance for modifying its properties, such as  $\text{SiC}$ ,  $\text{BeO}$ , heat resistant silicides other than  $\text{MoSi}_2$ , borides, and aluminides.

The additional constituent consists prefer-

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ably of a vitreous substance having a melting point lower than  $1720^{\circ}\text{C}$ . Suitable vitreous substances are those which contain between 60% and 100%  $\text{SiO}_2$  and a balance of one or more oxides of the following elements: Na, K, Ca, Mg, B, Al and Ba. The vitreous substance may thus consist of solely  $\text{SiO}_2$  having a melting point of about  $1720^{\circ}\text{C}$ .

A resistance element as manufactured in accordance with the invention may comprise a rod-like incandescent-zone body, or heating section, and, butt welded thereto, rod-like terminal-end conductors or terminal sections, of a cross-sectional area larger than that of the heating section but reduced at the welded joint to substantially the area of the heating section, the area increase caused by the upset of the weld being less than 40% of the cross-sectional area of the heating section of the element.

In the production of a resistance element according to the invention one starts with at least two sintered form bodies. According to the invention these bodies are joined or integrated, by butt welding by the combined steps of composing the bodies, at least, within the zones thereof influenced by the welding operation, of a basic substance of electrically conductive crystal grains which, *per se*, are lacking or have an insufficient degree of plasticity at the butt welding temperature, and an additional constituent uniformly distributed among the grains of the basic substance and having a lower softening temperature than that of the basic substance and make the body plastically deformable and ductile at the butt welding temperature, and, after sintering the form bodies into an approximately non-porous state, forcing the form bodies against each other while heating the same by electric current flowing through seld bodies.

Obviously, it would not be necessary that all of the grains within the basic substance be electrically conductive; it has been found, however, that the volume of conductive grains must exceed 20% by volume of the combined material to cause the latter to be electrically conductive. In this case the additional constituent should consist of components of such a nature that, when heated during the butt welding operation, they will form a vitreous material having a low melting point, preferably lower than  $1500^{\circ}\text{C}$ . The additional constituent has the effect of causing the material—at least in the welding zone—to exhibit increased ductility and plastic deformability in the butt welding operation. The grain size of the basic substance also increases so as to become at most 20 micra within the finished welding zone whereas the same is at most 10 micra outside this zone. When the butt welded joint is completed, and the element is subsequently subjected to heating during a fairly long time, a certain ageing will take place which reduces the deformability and ductility of the material.

The invention will now be described more in detail with reference to the accompanying drawings, in which:—

Fig. 1 is a diagrammatic showing of an electric resistance element manufactured in accordance with the invention,

Fig. 2 a fragmentary view of two bodies to be joined by butt welding,

Fig. 3 an illustration of various details of a butt welded joint,

Fig. 4 a diagrammatic showing of a butt welding apparatus,

Fig. 5 a cross-section through a deficient butt welded joint,

Fig. 6 a graphic representation,

Fig. 7 an illustration of a detail arrangement used in preparing the form bodies to be butt welded,

Figs. 8 and 9 a showing of two stages in the preparation of the terminal end portion of a resistance element,

Figs. 10 to 12 an illustration of the butt welding operation, and

Figs. 13 and 14 a showing of various steps of a different embodiment of the butt welding operation.

The initial materials, or material masses, which primarily may come into consideration for the formation of the various parts or sections of the resistance element, and which will be discussed hereinafter, can be classified as belonging to any one of the three following types:

I. A powder mixture consisting of 95%  $\text{MoSi}_2$  plus 5% by weight of the additional constituent. The latter may consist of 25%  $\text{Al}_2\text{O}_3$ , 65%  $\text{SiO}_2$  and a vitrifying substance, such as one or more oxides of the elements Na, K, Ca, Mg, B, Al and Ba. By sintering these materials an electrically conductive body will result.

II. 80% by weight of  $\text{MoSi}_2$ , 15%  $\text{Al}_2\text{O}_3$ , and 5% of the additional constituent having a composition as stated under I above. The sintered body produced from this powder mixture has an electrical resistivity substantially twice that of a body produced from the material according to point I above.

III. 60%  $\text{MoSi}_2$ , 35%  $\text{Al}_2\text{O}_3$ , and 5% of the additional constituent as stated under I. The body produced from this mixture has an electrical resistivity about four times that of the body produced from the material according to I.

From these materials, it is preferred to use either of Types I and II for producing the terminal end sections of the resistance element, whereas all three types may be used as a material for producing the heating section. However, as regards the welding operation specifically, the only portion of the bodies which is of importance to this operation is that which is influenced thereby within the so-called weld zone.

The resistance element shown in Fig. 1

comprises two terminal end sections  $A_1$ ,  $A_2$  having a diameter of, for instance, 13 mm. and of which the portions  $C_1$ ,  $C_2$  facing the heating section proper B are tapered towards and united with this heating section B by butt welded joints  $F_1$  and  $F_2$ . As diagrammatically illustrated in the drawing, the portions about each welded joint have been caused to expand somewhat radially due to the upsetting action, and the associated welding zones  $G_1$ ,  $G_2$  substantially comprise those upset portions. For example, with a diameter of the heating section B of 6 mm., the axial length of each such weld zone may be estimated to range between 5 and 10 mm.

Fig. 2 shows more in detail the formation of the junction portion  $C_1$ . The latter may suitably be mechanically reduced along a distance H of, for instance, 20 mm. and a radius R of, for instance, 70 mm. so as to have a diameter  $d_1$  which should be substantially equal to the diameter  $d_2$  of the heating section. However, these diameters should not differ from each other by an amount greater than that which is determined by the condition that, before the welding operation, the ratio of the cross-sectional area of the terminal end section to that of the heating section should range between 0.94 and 1.04. As a consequence of the upsetting action, as shown in Fig. 3 the diameter of the welded zone will be increased to a value  $d_3$ , and if the butt welding operation has taken place properly, the maximum area increase of the welded zone should range between 25% and 40%.

Fig. 4 illustrates apparatus for carrying out the butt welding operation, said apparatus comprising two clamping jaws 1 and 2 and, between these jaws, electric contact elements 3 and 4 adapted to be supplied with current through cables 5 and 6. A glove or socket 7 is provided with a supply conduit 8 for a neutral gas, such as argon, for example. The terminal end section  $A_1$  is adjusted into alignment with the heating section B so as to cause their ends  $G_1$  and  $G_2$  to meet within the glove or socket 7.

It is of importance that, before the welding operation, the abutting faces  $E_1$  and  $E_2$  should be ground to perfectly plane in order to obtain a maximum uniformity of upsetting in the compression. Further, before the upsetting operation, the welding zones  $G_1$  and  $G_2$  should have a constant diameter along a length of about 10 mm. from the abutting faces  $E_1$  and  $E_2$ . Practically suitable dimensions are, for example,  $d_1 = d_2 = 6$  mm., and  $D_1 = 13$  mm. The ratio of the electrical resistances per unit length of the sections  $A_1$  and B thus will be  $6^2/13^2$ , i.e. equal to 0.21. In the present instance the terminal end sections and the heating section may be made of one and the same material.

During the welding operation the welding current is controlled by means of an adjustable

transformer. It is of importance that a shielding gas flow of a suitable rate be maintained, and this rate of gas flow should be sufficient to displace the ambient air from the welding region, however, without being enough heavy to cause turbulence within the glove or socket 7 which might incur that the gas-stream would induce ambient air from the sides. In the apparatus shown and used a rate of gas flow of 14 to 15 litres per minute has been found to be most suitable. It is to be noted that in this respect that if air is given access to the welding surfaces, the adhesion of the welded joint will as a rule, be deficient. The pressure applied between the parts to be welded together is initially relatively moderate and is increased after the material has become ductile over the whole cross-sectional area. This increased pressure is maintained during 30 to 60 seconds. After the degree of upset has reached a suitable value, as stated above, the welding current is switched off while maintaining the final upsetting mechanical pressure. It should be observed here, that the degree of upsetting is critical and if the deformation is too great internal cracks or fissures within the upset portion may occur which, although they would not reach the external surface, nevertheless would deteriorate the mechanical strength of the weld. The degree of upsetting being too small this may result in remaining thin films of oxides or phases of a lower content of silicon than that of  $MoSi_2$ . The presence of such intermediate layers with characteristics differing from those of the basic material, such as a different coefficient of thermal expansion, will of course contribute to a weakening of the welded joint. With a normal degree of upsetting action, giving rise to a maximum area increase at the welded joint ranging between 25% and 40% on the other hand, a re-baking or interdiffusion will take place causing such films to be shattered and admixed with the basic material. If, further, the upsetting is performed too rapidly, this will result in the same kind of internal cracking as was described with reference to the case of the degree of the upsetting being too high.

The area increase involved in the welding operation is of advantage from the point of view of strength, in that the mechanical strength at the welded joint will be increased, the welding region being otherwise always somewhat weaker than the remainder of the element, in spite of the attainment of a sound weld with good interdiffusion.

It has been found that the result of the butt welding is highly dependent on the course of the preceding sintering of the bodies to be welded as regards the height and duration of the temperature applied in the sintering.

It is important, therefore, that, before the butt welding operation, the welding zones, at least, of the form bodies are sintered during a

length of time and at a temperature which are on one hand, sufficient to render the compacts approximately non-porous, but are, on the other hand, delimited in such a way that, within the associated welding zone, at least, the material will maintain after its sintering a plastic deformability while heated during a length of time sufficient for enabling the butt welding operation to be completed. It may be mentioned for guidance, that it has been found with an initial material which contains at least, within the intended welding zone, at least about 90%  $\text{MoSi}_2$  and an oxide component which may be expected in the finished bodies to consist of a glass containing  $\text{SiO}_2$ ,  $\text{Al}_2\text{P}$  and a fluxing agent and forming a content of 5 to 10% by weight of the material, that the compacts should be sintered in hydrogen gas for a time of up to 10 hours at a temperature ranging between  $950^\circ\text{C}$  and  $1450^\circ\text{C}$ , and then should be sintered in air for one hour, at most, at a surface temperature of  $1500^\circ\text{C}$  to  $1650^\circ\text{C}$ . The actual time and temperature to be employed within these limits should be selected in relation to the diameter or cross-sectional area of the bodies or compacts. As a matter of fact, if this diameter is less than approximately 10 mm, for instance 6 mm, which is usual value for the heating section, then the suitability of the body for butt welding purposes would not materially change as a result of a certain variation of the sintering time and temperature within the above-stated limit values. However, as regards the terminal end sections, the diameter of which will frequently amount to more than 10 mm., for instance 13 mm., conditions will be more critical, and one will be confined to a more restricted range, thus, with bodies of a kind containing at least 80%  $\text{MoSi}_2$ , the material of their welding zones, at least should be sintered at a temperature of  $1320^\circ\text{C}$  in hydrogen gas for a time of 15 to 20 minutes, and then be subjected to final sintering in air by passing an electric current therethrough during 5 minutes at a surface temperature of  $1500^\circ\text{C}$ .

Fig. 5 illustrates the surface of fracture of a weld which has been made by subjecting the material of a terminal end section to an excessive temperature for an excessive length of time before welding. The dotted surface 9 thereof corresponds to an ordinary surface of fracture where the weld is sound, whereas the surrounding sectioned surface 10 represents a portion where the welding action has not been successful, and the original ground end face is exposed in its initial condition.

Thus it is of the utmost importance, that the sintering process during the final sintering procedure be controlled which is feasible since this final sintering is carried out by the aid of an electric current passed through the body or compact. Fig. 6 illustrates a graph in which the ordinates represent electrical resistivity  $r$

and the abscissae represent the time  $t$  of a sintering process. With the lapse of time, the sintering action causes the pores of the material to be more and more reduced whereby the conductivity increases and the resistivity  $r$ , accordingly, diminishes. As an approximate point 11 is reached, the material will be almost non-porous, and the sintering should then be interrupted at once and the material allowed to cool down preparatory to the butt welding operation. As a matter of fact, if the final sintering be continued beyond this point certain ageing phenomena will occur causing the deformability or ductility of the material to gradually diminish, so that, as stated hereinbefore, the butt welding operation will be gradually more difficult to perform.

Where the production of an element is concerned of which the terminal end portions are larger in diameter than the heating section, as shown in Figs. 1, 2 and 3, it is particularly important to observe that the welding zones  $G_1$  of the terminal end sections are made of a material which can be expected, after the welding operation, to possess a coefficient of thermal expansion substantially the same as that of the material of the mating welding zones  $G_2$  of the heating section, and that the terminal end sections be sintered into a substantial non-porous state, and be shaped, suitably by mechanical working, into a cross-sectional area substantially equal to that of the heating section, and that the terminal end sections are butt welded to the heating section. The mechanically worked portions of the terminal end sections due to the grinding thereof, will lack the protective skin or coating of quartz glass, formed on sintered bodies constituted to their major portion of  $\text{MoSi}_2$ . Therefore, these portions should be subjected to sintering subsequent to the butt welding operation so as to obtain such a protective coating which will, as a rule, be obtained after a short time of service use of the resistance element at a high temperature.

As the protective coating is formed only if the oxidizing sintering takes place at about  $1200^\circ\text{C}$  or higher and, if this temperature is to be obtained at the ground surfaces of the cool terminal sections through heat conduction from the hot heating section, the grinding must be made in such a way that the desired temperature is obtained on the ground surfaces. The grinding should thus be made along a distance of at most 25 mm. on the terminal section.

It is previously known *per se* to produce the resistance elements powder-metallurgically by sintering the same from two different powder masses. When such sintered bodies are to be used as terminal sections for the purpose here contemplated, the welding zones of the bodies facing the heating zone should have a composition similar to that of the welding zones of the heating section (or incandescent

zone) after which the bodies or compacts forming the terminal end sections and heating section are sintered individually by passing electric current therethrough, and the welding zones of the terminal end sections and heating section are then butt welded by means of electric current.

This procedure can be carried into effect in different ways Fig. 8 illustrates one such method by way of example. In this case, the two terminal end bodies are compacted from two different powder masses, such as of Type I and II, that of Type II being intended to form the welding zones G, and that of Type I the cold zones A<sub>3</sub> and A<sub>4</sub>, so as to produce an integral, or unitary form body or compact A<sub>3</sub>-G-A<sub>4</sub> form a terminal section blank. Thus the cross-sectional area of this blank will vary along the length of the blank in such a way in dependence of the resistivity of the material present at the cross-section concerned that the desired sintering temperature will obtain. This blank compact may be finally sintered by applying two electrodes 12, 13 thereto and connecting these electrodes to a source of electric current (not shown) through an ammeter 14. The passage of the current will cause the body to be heated approximately uniformly, or according to a desired distribution, and the sintering is continued until it is ascertained, by the reading of the ammeter 14, that a nearly non-porous state has been attained, after which the intermediate portion G (see Fig. 9) is cut off into two welding zones G<sub>3</sub> and G<sub>4</sub> which (see Fig. 10) are reduced by grinding so as to form the narrow welding zone g<sub>3</sub> having a diameter substantially equal to that of the heating section B. Butt welding can then be undertaken at 15 by passing an electric resistance heating current therethrough. After completing the welding operation proper the material has been subjected to an upsetting pressure causing an upset or bead 16 to form about the welding joint.

The procedure could instead be carried out in the following manner, as illustrated in Fig. 13. The form body there shown consists of a reduced intermediate portion A<sub>5</sub>-A<sub>6</sub> made of a terminal end section material, such as of Type I, and two relatively enlarged end portions G<sub>5</sub>-G<sub>6</sub> made of a heating section material, such as of Type II. This form body is initially produced by compacting the powder mixtures and sintering the compact at a temperature which enables the compact to be handled. Thereafter preferably water-cooled contacts are connected at 17 and 18, and sintering is undertaken in a shielding atmosphere at a sufficiently high temperature to render the compact low-porous and mechanically durable except at its extreme ends. If the heating section material at G<sub>5</sub> and G<sub>6</sub> is to have a sintering temperature higher than that of the terminal section material at A<sub>5</sub> and A<sub>6</sub> then the ratio of the cross-sectional area of the in-

termediate portion A<sub>5</sub>-A<sub>6</sub> to that of the end portions G<sub>5</sub> and G<sub>6</sub> should exceed the ratio of the electric resistivity of the heating section material to that of the terminal section material. If, conversely, the sintering temperature of the terminal section material is to be higher, the reverse condition applies. After completing the final sintering operation, the form body is cut off at 19 into two equal parts, and the dotted regions at the ends are ground away.

In connection with the production of the form body G<sub>5</sub>-A<sub>5</sub>-A<sub>6</sub>-G<sub>6</sub> it may be desirable to establish diffuse or continuous (or interfused) junctions between the heating section material in G<sub>5</sub> and G<sub>6</sub> and the terminal section material in A<sub>5</sub>, A<sub>6</sub>, these interfused junctions extending along predetermined lengths 20 and 21, respectively. This feature will be particularly advantageous and desirable where the difference in sintering temperatures between the heating section material and the terminal section material is great. The terminal section part G<sub>5</sub>-A<sub>5</sub> could possibly be combined with an extended terminal end section 22 and a heating section B in a manner corresponding to that described hereinbefore. The butt welding operation can readily take place since the material on either sides of the welding localities 23 and 24 is the same, or substantially the same. The remaining enlargement or upset 25 will cause the high temperature created in the heating section B to diminish sufficiently not to destroy the terminal section material A<sub>5</sub> in the event that the latter should have a resistance to heat lower than that of the heating section material.

In the manner above described it is thus feasible to produce elements having different compositions in heating section and terminal sections. The cooler portions of the terminal sections may be advantageously made in metal which often is cheaper and more ductile than, for instance, MoSi<sub>2</sub>. Such a metal may be a known heat resistant oxidation proof alloy or, for instance, molybdenum protected by surface siliciding. It is also possible to produce combinations of SiC and MoSi<sub>2</sub> for the heating zone and the terminal sections in resistance elements according to the present invention.

It may be mentioned by way of example that the blank of Fig. 13 can be formed by press-moulding from the two powder mixtures into the following dimensions: a length of 500 mm. and a cross section which is 6 by 6 mm. throughout the reduced intermediate portion and 6 by 15 mm. along the enlarged end portions. The length of the intermediate portion is 2 by 210 mm., and the length of each end portion is 40 mm. The heating section rod B may be an extruded rod of circular section having a length of 1200 mm. and a diameter of 6 mm.

It is possible also to integrate sintered bodies of differently composed materials di-

rectly by butt welding, provided that after the welding operation they will exhibit equal coefficients of thermal expansion. On the other hand, they may exhibit different electric resistivities. Therefore, in connection with the butt welding operation one provides for separate cooling of the welding zone possessing the higher electrical resistivity value, i.e. for the heating section, as a rule, thereby to render the two welding zones equal in softness, or ductility, so as to be upset to equal degrees as a result of the pressure application in the butt welding operation. Such separate cooling, as shown in Fig. 7, may be effected by directing a blast of an inert gas, for instance argon, onto the welding zone 25 of the heating section, whereas the welding zone 26 of the terminal end section is not subjected to such cooling. The cooling medium can be supplied through separate jets 27 and 28 which are introduced into the interior of the socket or glove 7 shown in Fig. 4.

Finally, applicant has found that the vitrifying substance to be incorporated in or to constitute the additional constituent is preferably a finely divided clay of the montmorillonite group. In producing the sintered body the following steps are preferably taken:

Finely divided molybdenum disilicide is admixed with about 0.2 to 20% by weight of the finely divided plastic clay, the mixture sintered in the absence of oxygen at about 1000 to 1400° C, thereby producing a pre-sintered porous material, and then the sintered material is heated in the presence of oxygen at a temperature of 1400 to 1700° C at which the clay is melted whereby the resistance of the sintered body to oxidation, grain growth, creep and recrystallization is enhanced.

#### WHAT WE CLAIM IS:—

1. An electric resistance element, composed of at least two powder-metallurgically produced sintered bodies united with each other and each having a content of at least 20% by volume of electrically conductive particles, characterized in that said bodies are joined together by an electric resistance butt welding operation and are built-up, at least within the zone thereof influenced by the welding operation, in part by a basic substance of crystal grains, which substance, *per se*, lacks or has insufficient plastic deformability and ductility during the butt welding operation, and in part by an additional constituent uniformly distributed in the basic substance and having a softening temperature lower than that of the basic substance, the grain size of the basic substance being at most 20 micra in the welding zone and at most 10 micra beyond this zone.

2. A resistance element according to claim 1, characterized in that the basic substance forms at least 20% and at most 99% by volume of the material of the welding zone.

3. A resistance element according to claim 1 or 2, characterized in that the major pro-

portion of the basic substance, preferably more than 65% by volume thereof, consists of MoSi<sub>2</sub>.

4. A resistance element according to claim 1 or 2, characterized in that the basic substance consists of at least 25% by volume of MoSi<sub>2</sub> and at most 75% by volume of Al<sub>2</sub>O<sub>3</sub>.

5. A resistance element according to claim 1 or 2, characterized in that the additional constituent constitutes from 5 to 25% by volume of the material in the weld zone.

6. A resistance element according to any one of the claims 1 to 5, characterized in that the additional constituent consists of a vitreous substance (glass) having a melting point lower than 1720° C.

7. A resistance element according to any of the claims 1 to 6, characterized in that the element comprises a rod-like incandescent-zone body, or heating section, and, butt welded thereto, rod-like terminal-end conductors, or terminal sections, of a cross sectional area larger than that of said heating section but reduced at the welded joint to substantially the area of said heating section, the area increase caused by the upset of the weld being less than 40% of the area of the heating section of the element.

8. A method of manufacturing the resistance element claimed in any one of the preceding claims 1 to 7, starting with at least two powder-metallurgically produced form bodies, characterized in that the form bodies are joined together by butt welding by the combined steps of composing the bodies, within the zones thereof influenced by the welding operation, at least, of a basic substance of electrically conductive crystal grains which, *per se*, are lacking, or have an insufficient degree of, plasticity at the butt welding temperature, and an additional constituent uniformly distributed among the grains of the basic substance and having a softening temperature lower than that of the basic substance and, after sintering the form bodies (or compacts) into an approximately non-porous state, forcing the form bodies against each other while heating the same by passing an electric current through said bodies.

9. A method according to claim 8, characterized in that said additional constituent consists of additions of a nature to form, when heated, a vitreous material (glass) having a low melting point (lower than 1500° C).

10. A method according to claim 8 or 9 of manufacturing an electric resistance element comprising an intermediate heating section and terminal sections welded thereto, characterized in that, before the butt welding operation, the welding zones, at least, of the form bodies (or compacts) are sintered during a length of time and at a temperature which are, on one hand, sufficient to render the compacts approximately non-porous, but are, on the other hand, delimited in such a way that,

within the associated welding zone; at least, the material will maintain after its sintering a plastic deformability while heated during a length of time sufficient for carrying out the butt welding operation.

11. A method according to claim 10, characterized in that the initial material contains, within the intended welding zone, at least, not less than 80%  $\text{MoSi}_2$  and an oxide component which may be expected in the finished bodies to consist of a glass containing  $\text{SiO}_2$ , aluminium oxide and a fluxing agent and forming a content of 5% to 10% by weight of the material; in that the compacts are sintered in hydrogen gas for a time of up to 10 hours at a temperature ranging between  $950^\circ\text{C}$  and  $1450^\circ\text{C}$  and are then sintered in air for 1 hour, at most, at a surface temperature of  $1500^\circ\text{C}$  to  $1650^\circ\text{C}$ .

12. A method according to claim 11, characterized in that the material of the welding zones, at least, is sintered at a temperature of  $1320^\circ\text{C}$  in hydrogen gas for a time of 15 to 20 minutes, and is then subjected to final sintering in air by passing an electric current therethrough during 5 minutes at a surface temperature of  $1500^\circ\text{C}$ .

13. A method according to any one of the claims 8 to 12 of producing an element having terminal sections which are enlarged relative to its heating section, characterized in that the welding zones of said terminal sections consist of a material which may be expected after the welding operation to exhibit a coefficient of thermal expansion substantially the same as that of the material of the mating welding zones of the heating section; and are sintered into an approximately non-porous state and are shaped, preferably by mechanical working, into a cross-section substantially equal to that of the welding zones of the heating section, after which they are butt welded to these welding zones.

14. A method according to claim 13, characterized in that the mechanically worked portions of the terminal section bodies are heated in an oxidizing atmosphere after the butt welding operation so as to cause the formation of a protective coating of  $\text{SiO}_2$  thereon through oxidation of  $\text{MoSi}_2$  present in the body.

15. A method according to claim 13 or 14 characterized in that, before the welding operation, the ratio of the cross-sectional area of the welding zones of the terminal sections to that of the welding zones of the heating section ranges between 0.94 and 1.04.

16. A method according to any one of the claims 8 to 15, characterized in that the area increase of the cross-sections of the welding zones due to the upset formed by the butt welding ranges between 25% and 40%.

17. A method according to the claims 8 to 16, characterized by carrying out the butt

welding in a shielding gas, such as argon, for example.

18. A method according to any one of the claims 8 to 17, characterized in that the terminal section bodies are produced powder-metallurgically by sintering compacts of two different powder masses so natured as to cause their welding zones facing the heating section body to be of a composition similar to that of the mating welding zones of the heating section, after which the terminal section bodies and the heating section body are sintered individually by passing electric current therethrough, and the mating welding zones of the terminal sections and the heating section are butt welded by passing electric current therethrough.

19. A method according to claim 18, characterized in that the two terminal section bodies are compacted from two different powder masses of which one is intended to form the welding zones of one single integral form body forming a terminal section blank, the intermediate portion of which is produced from one powder mass whereas its end portions are produced from the other powder mass, and the cross-sectional area of which varies along the blank in such a way depending on the electrical resistivity of the material present at the cross-section concerned that the desired sintering temperature will obtain.

20. A method according to claim 19, characterized in that the powder masses are so distributed in the terminal section blank as to cause the intermediate portion thereof to form a welding zone having a composition corresponding to that of the heating section body and having a cross-sectional area larger than its end portions, and in that the terminal section blank is sintered into a nearly non-porous state by passing electric current therethrough, after which said intermediate portion is cut off into two welding zones and is reduced by grinding into a cross-sectional area substantially the same as that of the mating welding zones of the heating section body and is joined with the latter by butt welding.

21. A method according to claim 19, characterized in that the powder masses are so distributed in the terminal section blank as to cause the intermediate portion of the latter to be of a composition corresponding to that of the ends of the terminal section body remote from the heating section body, and to have a cross-sectional area smaller than that of its welding-zone forming end portions, and in that the terminal section blank is sintered into a nearly non-porous state by passing electric current therethrough, after which said intermediate portion is cut off and its welding zones are reduced by grinding into a cross-sectional area substantially the same as that of the mating welding zones of the heating section body, and is joined with the latter by

butt welding.

- 5 22. A method according to any one of the claims 8 to 17, characterized in that welding zones produced in form bodies of different composition but substantially equal in cross-sectional area, are joined together by a butt welding operation in which operation the bodies are subjected to different cooling actions so as to obtain substantially equal softness, or ductility, characteristics.
- 10 23. A method according to any one of the claims 11 to 22, characterized in that said additional constituent in the initial material

is bentonite or other clays related thereto.

24. Electric resistance element constructed and arranged substantially as described herein with reference to the accompanying drawings. 15
25. A method of manufacturing an electric resistance element substantially as described herein with reference to the accompanying drawings. 20

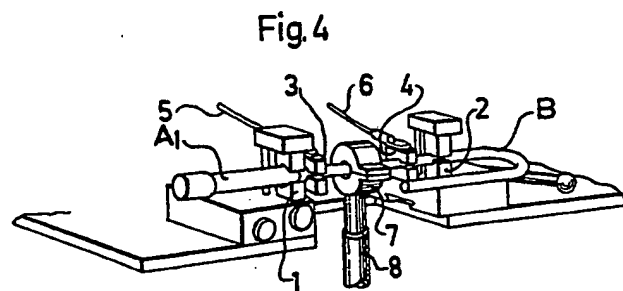
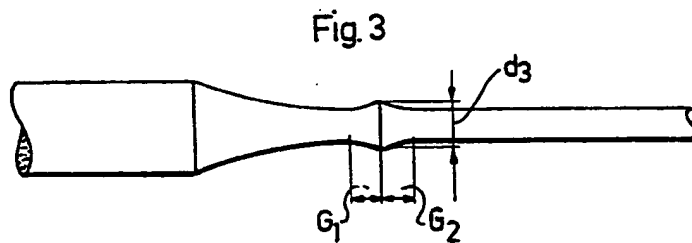
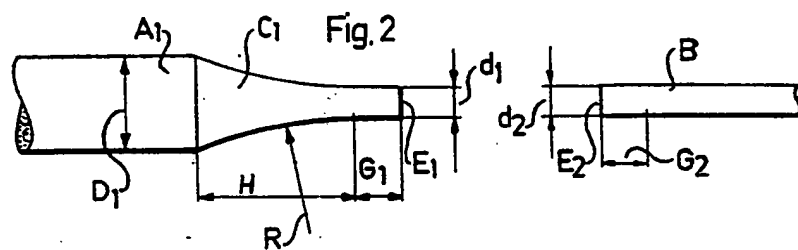
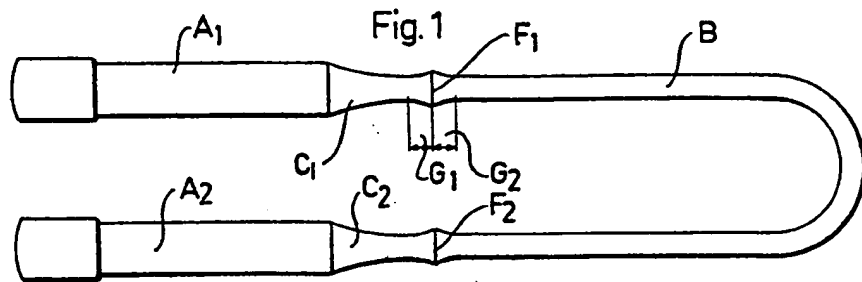
For the Applicants,

F. J. CLEVELAND & COMPANY,  
Chartered Patent Agents,

29, Southampton Buildings, Chancery Lane,  
London, W.C.2.

Leamington Spa: Printed for Her Majesty's Stationery Office, by the Courier Press.—1960.  
Published by The Patent Office, 25, Southampton Buildings, London, W.C.2, from which  
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834,739 COMPLETE SPECIFICATION

4 SHEETS

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SHEETS 1 & 2

Fig. 5

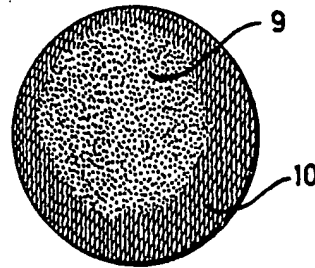


Fig. 6

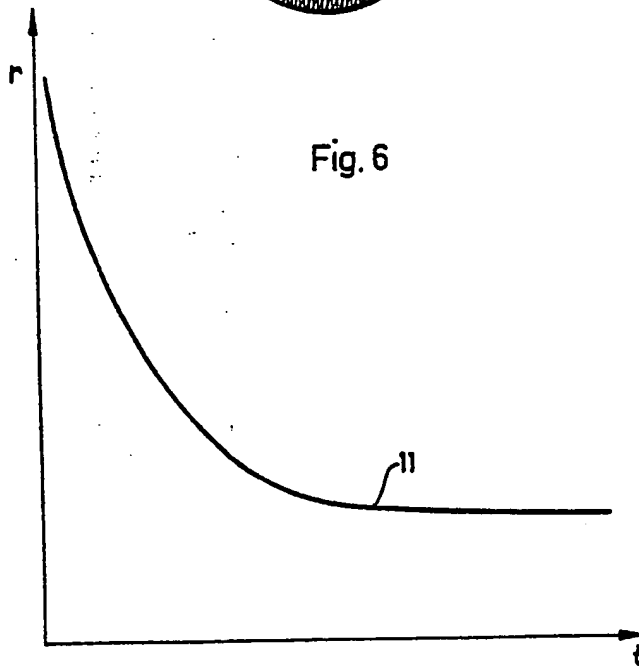
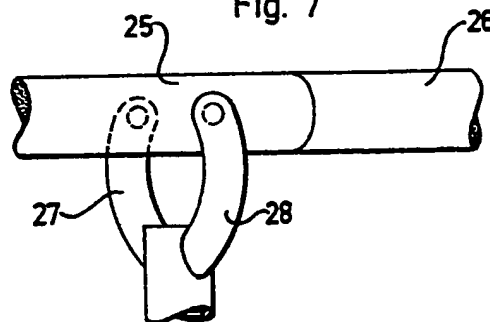


Fig. 7



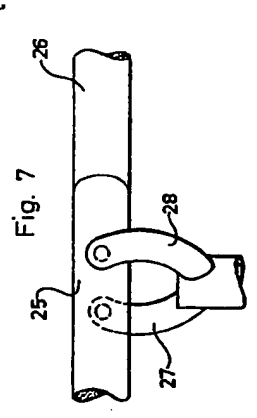
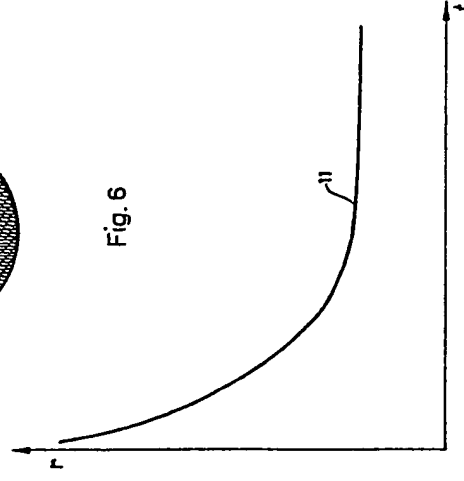
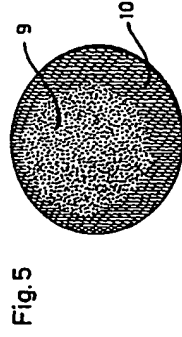
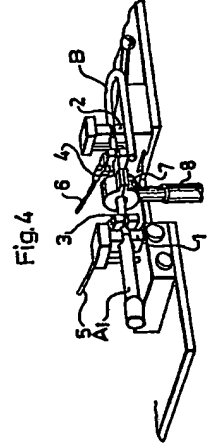
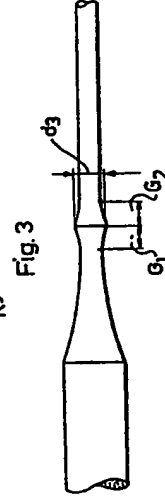
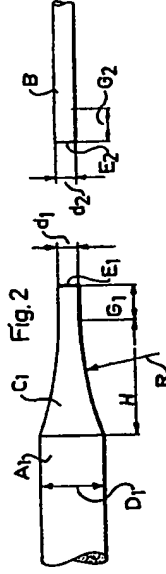
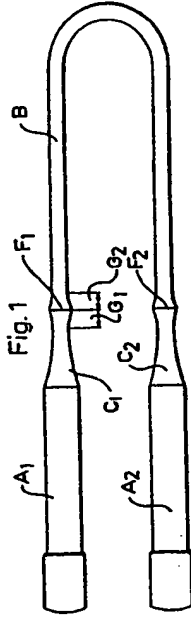


Fig. 8

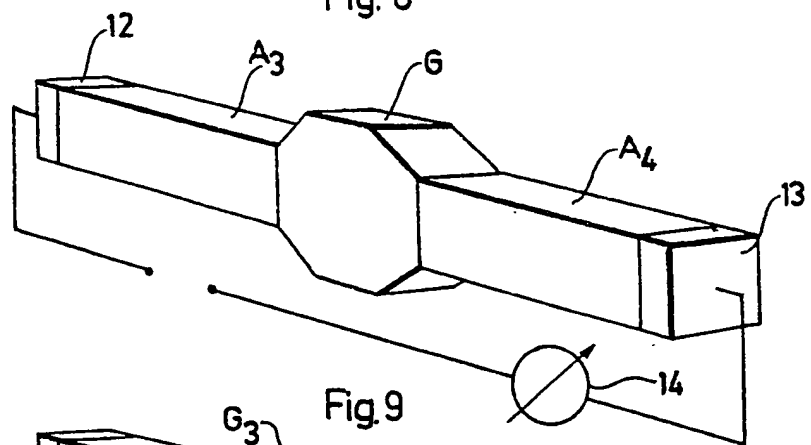


Fig. 9

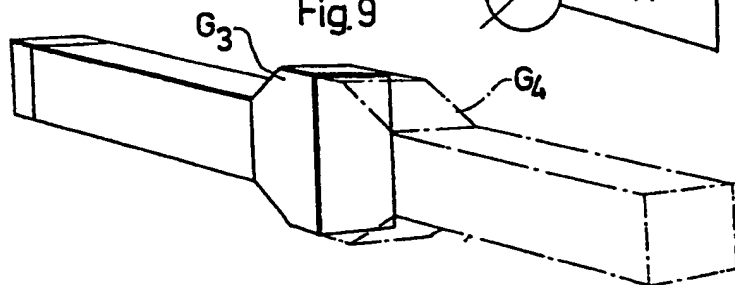


Fig. 10

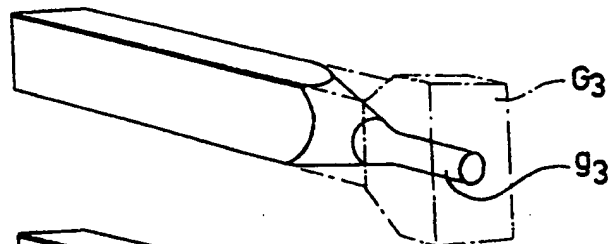


Fig. 11

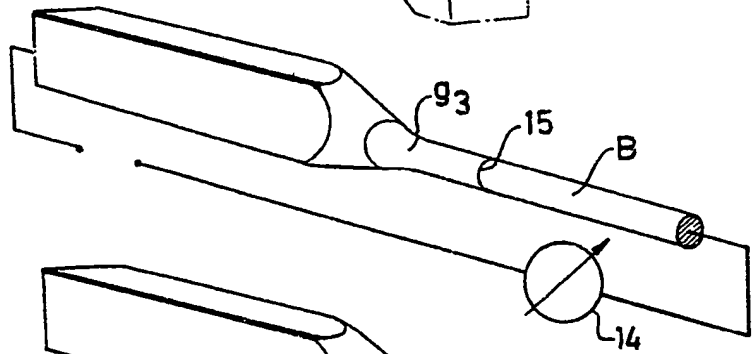
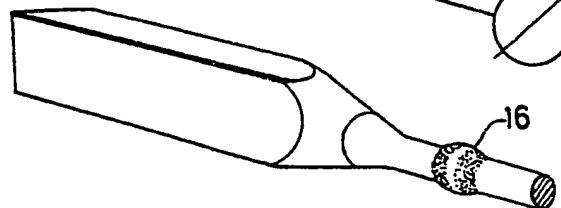


Fig. 12



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SHEETS 3 & 4

Fig.13

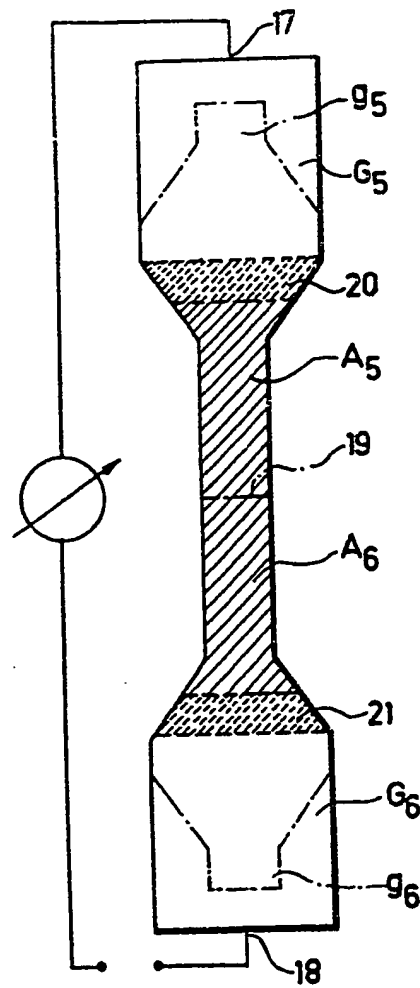
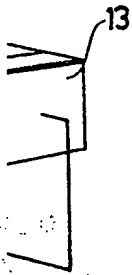
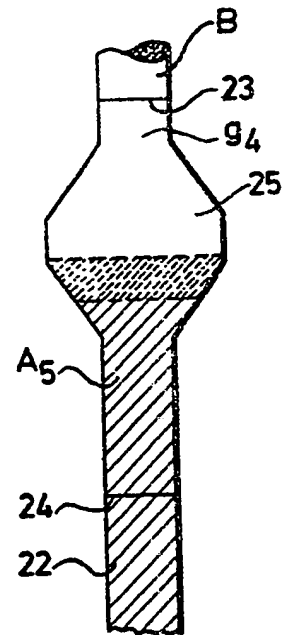
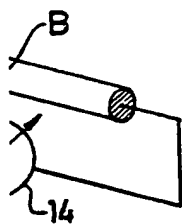


Fig.14



G3

-G3



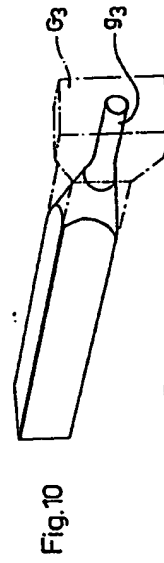
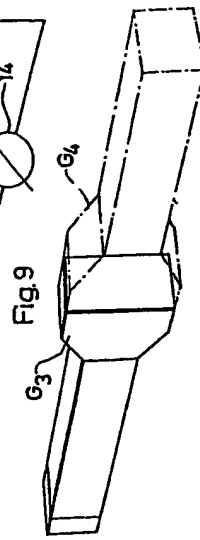
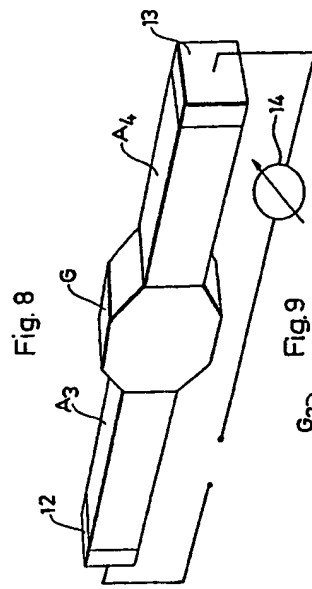


Fig. 13

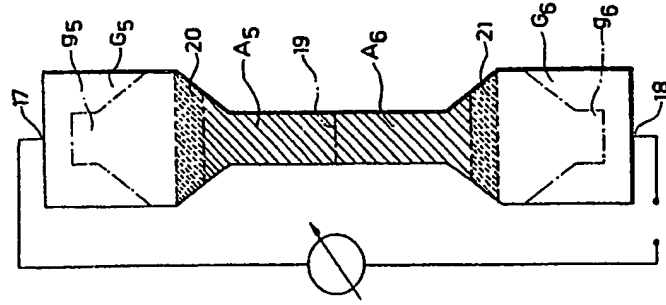
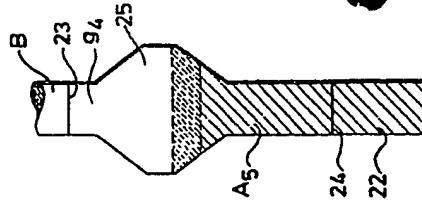


Fig. 14



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